

# Probing equilibrium of molecular and deprotonated water on TiO<sub>2</sub>(110)

Zhi-Tao Wang<sup>a,b,1</sup>, Yang-Gang Wang<sup>b,c,1</sup>, Rentao Mu<sup>b,c,1</sup>, Yeohon Yoon<sup>b,c</sup>, Arjun Dahal<sup>a,b</sup>, Gregory K. Schenter<sup>b,d</sup>, Vassiliki-Alexandra Glezakou<sup>b,c</sup>, Roger Rousseau<sup>b,c,2</sup>, Igor Lyubintsky<sup>b,c,2,3</sup>, and Zdenek Dohnalek<sup>b,c,2</sup>

<sup>a</sup>Microscopy Group, Environmental Molecular Sciences Laboratory, Pacific Northwest National Laboratory, Richland, WA 99352; <sup>b</sup>Institute for Integrated Catalysis, Pacific Northwest National Laboratory, Richland, WA 99352; <sup>c</sup>Catalysis Group, Physical Sciences Division, Physical and Computational Sciences Directorate, Pacific Northwest National Laboratory, Richland, WA 99352; and <sup>d</sup>Chemical Physics & Analysis Group, Physical Sciences Division, Physical & Computational Sciences Directorate, Pacific Northwest National Laboratory, Richland, WA 99352

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**Understanding adsorbed water and its dissociation to surface hydroxyls on oxide surfaces is key to unraveling many physical and chemical processes, yet the barrier for its deprotonation has never been measured. In this study, we present direct evidence for water dissociation equilibrium on rutile-TiO<sub>2</sub>(110) by combining supersonic molecular beam, scanning tunneling microscopy (STM), and ab initio molecular dynamics. We measure the deprotonation/protonation barriers of 0.36 eV and find that molecularly bound water is preferred over the surface-bound hydroxyls by only 0.035 eV. We demonstrate that long-range electrostatic fields emanating from the oxide lead to steering and reorientation of the molecules approaching the surface, activating the O–H bonds and inducing deprotonation. The developed methodology for studying metastable reaction intermediates prepared with a high-energy molecular beam in the STM can be readily extended to other systems to clarify a wide range of important bond activation processes.**

adsorbate dynamics | water | dissociative adsorption | titanium dioxide | kinetic barriers

**W**ater is ubiquitous in the environment and, as such, the nature of its interactions with interfaces can determine the outcome of a broad range of processes that include wetting, dissolution, precipitation, phase transformation, corrosion, and catalytic and environmental reactions (1–7). In this regard, the relative stability of molecularly and dissociatively bound species can be of critical importance with the preferred configuration being controlled by many factors including surface structure, acid/base properties, defects, impurities, water coverage, and temperature (8–13). For oxides in particular, the relative stability of molecularly and dissociatively bound water has been widely debated even on the simplest, low-index surfaces.

Here we focus on resolving the fundamental question of water binding on rutile TiO<sub>2</sub>(110), one of the most studied oxide surfaces, which is often used as a prototype for reducible oxide surfaces and a model for understanding photocatalytic water splitting (3, 14–17). Interestingly, despite the overwhelming wealth of literature, the nature of water adsorption and dissociation on nondefect titanium sites has been disputed for decades and to date remains unsettled (3, 14–17). The underpinning difficulty in resolving this debate is that it is practically impossible to prepare stoichiometric TiO<sub>2</sub>(110) surfaces. As such, bridging hydroxyl groups formed by water dissociation in oxygen vacancy defects interfere with determining the extent of dissociation on regular Ti sites (3, 14–19). A number of recent studies by a variety of techniques including X-ray photoelectron spectroscopy (XPS) (20), infrared reflection absorption (21), photoelectron diffraction (PhD) (22), and scanning tunneling microscopy (STM) (23–25) arrived at conflicting conclusions. Whereas the XPS and PhD studies concluded partial dissociation of water in the hydrogen-bonded chains on Ti sites at higher coverages (20, 22), others are in favor of molecular bonding (21, 25).

To address the adsorption configuration of isolated water molecules, extremely low coverages are required. As such, the ensemble-averaged methods generally do not possess sufficient sensitivity and STM is the method of choice. Although the appearance of water-related features is suggestive of molecular binding, indirect evidence strongly indicates that water monomers can easily access the dissociated configuration (18, 19). For example, in the water-assisted diffusion of bridging hydroxyl hydrogen (18) and Ti-bound oxygen adatoms (19), water monomer dissociation represents a key step in the proposed mechanism. This is further supported by the wealth of theoretical studies that yield very close adsorption energies (within ~0.1 eV) for molecularly and dissociatively bound water monomers (14–18).

To address this topic, we have constructed a unique instrument that allows for in situ molecular beam scattering studies directly under the STM tip, an instrument combination that has hitherto proven difficult to achieve (26–28). In this study, we used hyperthermal water beams with variable energy and probed the probability of the formation of metastable binding configurations on TiO<sub>2</sub>(110) at low substrate temperatures (~80 K). On metals, such molecular beam studies have proven indispensable for the understanding of dynamic factors controlling dissociation processes such as the excitation of vibrational modes, energy flow, and steering (29–31). We demonstrate that the observed metastable configurations are composed of terminal and bridging

## Significance

**Understanding how water binds and dissociates on surfaces has broad implications in a vast range of physical and chemical processes. The relative stability of molecularly and dissociatively bound water has been debated for decades on many oxide surfaces, but it has never been successfully measured. Our study describes unique instrumentation, direct measurements, and a state-of-the-art computation and theory approach that yield a detailed kinetic and dynamic description of water deprotonation equilibrium on TiO<sub>2</sub>(110), a prototypical surface commonly used in mechanistic studies of photocatalytic water splitting. This unique study demonstrates that the molecularly bound water on TiO<sub>2</sub>(110) is preferred over the surface-bound hydroxyls by only 0.035 eV.**

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<sup>1</sup>Z.-T.W., Y.-G.W., and R.M. contributed equally to this work.

<sup>2</sup>To whom correspondence may be addressed. Email: roger.rousseau@pnnl.gov, igor.lyubintsky@oregonstate.edu, or zdenek.dohnalek@pnnl.gov.

<sup>3</sup>Present address: School of Chemical, Biological and Environmental Engineering, Oregon State University, Corvallis, OR 97331.

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hydroxyl pairs that can be converted back to molecularly bound water at higher temperatures (>100 K). In parallel, theoretical protocols were developed to reveal mechanistic details, guide experiments, and provide data interpretation. We accurately determined the equilibrium distribution and the interconversion energy barriers. The analysis reveals that the dissociated water configuration is only 0.035 eV higher in energy. A unique finding of this study is that long-range (5–10-Å) electrostatic coupling between the incident molecules and the oxide surface results in a strong steering and reorientation before collision.

## Results and Discussion

The STM images of clean  $\text{TiO}_2(110)$  composed of rows of topographically low-lying fivefold-coordinated titanium ions ( $\text{Ti}_{5c}$ , imaged bright) and high-lying bridging oxygen ions ( $\text{O}_b$ , imaged dark) are shown in Fig. 1 *A* and *C*. A small fraction of  $\text{O}_b$ s is missing, resulting in vacancies ( $\text{V}_o$ s) (15).

Fig. 1*B* shows the result after dosing  $\text{H}_2\text{O}$  with a room-temperature effusive molecular beam (incident energy of 0.06 eV) on the  $\text{TiO}_2(110)$  at 80 K. In all STM experiments,  $\text{H}_2\text{O}$  coverage was kept very low ( $\sim 0.05$  ML) to keep the observed molecules isolated. Only one type of feature, corresponding to water monomers centered on the  $\text{Ti}_{5c}$  rows (17, 18), is seen. Their uniformity is demonstrated in a high-contrast image (Fig. 1*B*, *Right*) and by the reproducibility of the line profiles (Fig. 1*B*, *Inset*, *Right*). Under limited surface mobility conditions (<160 K), the  $\text{H}_2\text{O}$  molecules do not diffuse on the  $\text{Ti}_{5c}$  rows on an experimentally observable timescale (24). As such, they are unable to reach  $\text{V}_o$  sites where they are known to dissociate and form bridging hydroxyl pairs ( $\text{HO}_b$ s) (3, 14, 15).

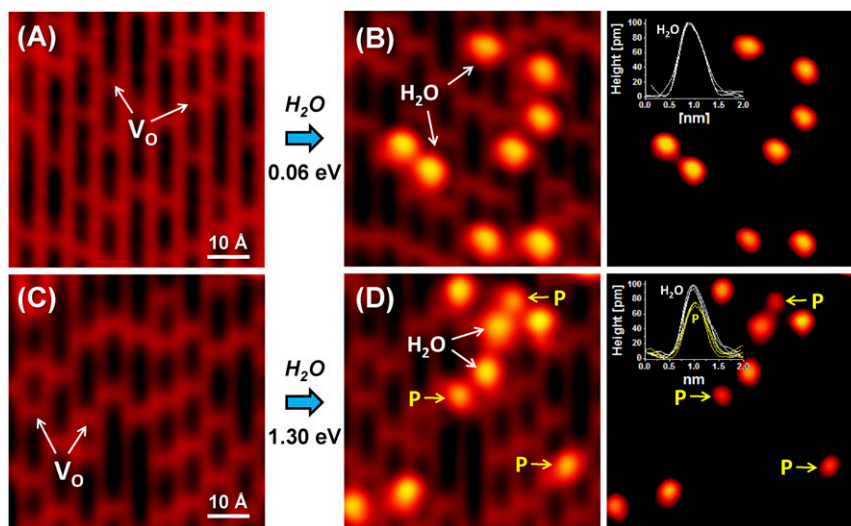
Whereas the  $\text{H}_2\text{O}$  features (Fig. 1*B*) are centered on  $\text{Ti}_{5c}$  sites, their appearance cannot be taken as conclusive evidence for the purely molecular binding of water. Under the equilibrium,  $\text{H}_2\text{O} + \text{O}_b \rightleftharpoons \text{HO}_t + \text{HO}_b$ , where  $\text{HO}_t$  is the  $\text{Ti}_{5c}$ -bound terminal hydroxyl, the majority species ( $\text{H}_2\text{O}$ ) determine the appearance. The evidence comes indirectly from studies of water reactions with O adatoms (19) and from water-assisted cross-row hydrogen transfer of  $\text{HO}_b$  species (18). Recent theoretical studies predict the higher stability of molecularly bound  $\text{H}_2\text{O}$  by less than 0.1 eV (13, 16–18).

We further probe the preparation of metastable species by colliding water molecules of high-incident energy (1.3 eV in Fig. 1*D*). In contrast to the low-incident energy experiment (Fig. 1*B*), two distinct types of features (65%  $\text{H}_2\text{O}$  and 35% P) are observed. The P features are round and also centered on  $\text{Ti}_{5c}$  sites. High contrast is needed (Fig. 1*D*, *Left*) to clearly discern their differences from  $\text{H}_2\text{O}$ . The line profiles in the inset further reveal that the P features are distinctly smaller and  $\sim 30\%$  less intense than  $\text{H}_2\text{O}$ . Whereas in the area shown in Fig. 1*B* the P features are not present, they comprise  $\sim 5\%$  of the total population at low  $\text{H}_2\text{O}$  incident energy. The large-scale images illustrating the overall distribution of the species at both low- and high-incident energies are shown in *SI Appendix*, section *S1* and Fig. *S1*.

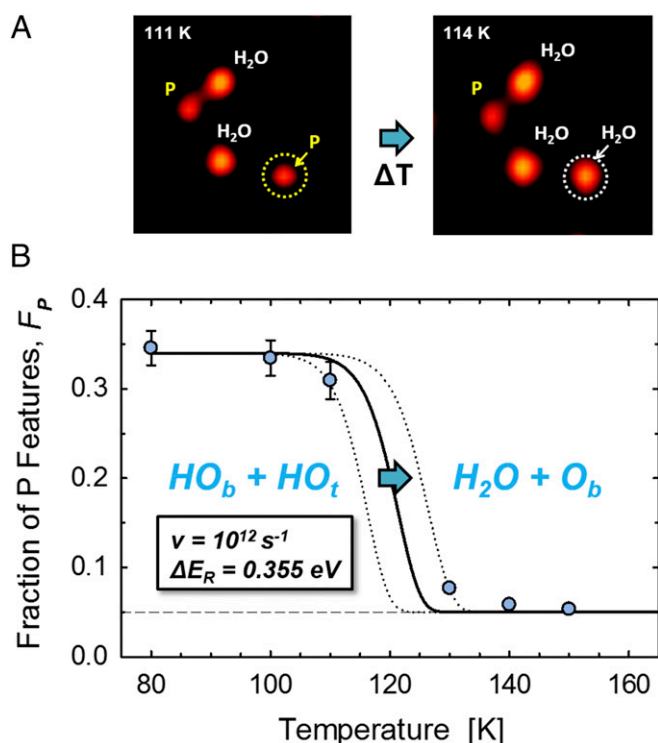
The appearance of the P features is surprising, as heuristically, pairs of two bright features ( $\text{HO}_t$  and  $\text{HO}_b$ ) due to water dissociation are expected. To further interrogate the chemical makeup of the P species, we carry out temperature-dependent experiments for water with incident energy of 1.3 eV (Fig. 2). These experiments allow us to follow the thermal equilibration of the  $\text{HO}_t$  and  $\text{HO}_b$  species with  $\text{H}_2\text{O}$ .

In the first type of experiment, we image continuously the same area while the sample temperature is increased from 111 to 114 K (Fig. 2*A*). The conversion of one P feature (yellow circle) to  $\text{H}_2\text{O}$  (white circle) shows that the P features contain all of the atoms from the water molecule and that none of them has been irreversibly lost to the vacuum or subsurface. This evidence strongly supports that the P features comprise the  $\text{HO}_b/\text{HO}_t$  pairs. Additional evidence from tip-pulse manipulation experiments and STM image simulations is provided in *SI Appendix*, section *S1* and Figs. *S2–S4*.

We quantify the temperature-dependent fraction of P features,  $F_P(T)$ , relative to the total number of  $\text{H}_2\text{O} + \text{P}$  features observed on the surface,  $F_P(T)$ , in Fig. 2*B* (circles). The initial  $\text{H}_2\text{O}$  adsorption temperature is stabilized between 80 and 150 K, water is dosed, and the surface is imaged. Below  $\sim 110$  K,  $F_P$  remains approximately constant and equal to 0.35. Above 110 K, a sharp drop is detected and  $T > 130$  K,  $F_P$  drops to 0.05 and remains approximately constant up to 150 K.



**Fig. 1.** Same area empty-state STM images before (*A* and *C*) and after (*B* and *D*) adsorption of 0.05 monolayer (ML) of water on  $\text{TiO}_2(110)$  at 80 K with the incident energy of 0.06 (*B*) and 1.30 eV (*D*). Images *A* and *C* show clean  $\text{TiO}_2(110)$  [ $\text{Ti}_{5c}$ , bright rows;  $\text{O}_b$ , dark rows;  $\text{V}_o$  defects (15%) appear as bright spots on dark  $\text{O}_b$  rows]. (*B*) A single type of features (labeled  $\text{H}_2\text{O}$ ) is observed on  $\text{Ti}_{5c}$  rows at low-incident energy. A high-contrast image and the inset with three different line profiles along the  $\text{Ti}_{5c}$  rows in *B* (*Right*) highlight their uniformity. (*D*) Adsorption of the high-incident energy water (1.30 eV; *D*, *Left*) shows two types of features,  $\text{H}_2\text{O}$  and features labeled P. High-contrast image (*D*, *Right*) and the *Inset* further highlight their differences. The angle of incidence of the molecular beam is fixed at a polar angle of  $60^\circ$  with respect to the surface normal and azimuth pointing across the rows.



**Fig. 2.** (A) Thermally induced conversion of P features to H<sub>2</sub>O following the adsorption of 0.05 ML of H<sub>2</sub>O with incident energy of 1.3 eV. STM images (Left,  $T = 111$  K; Right, 114 K) from an extended sequence obtained during the slow temperature ramp (0.1 K/min from 110 K). (B) The fraction of P features determined from the STM images obtained ~30 min after the H<sub>2</sub>O dose at different temperatures. The solid and dotted lines are the fits to the kinetic model. The error bars are determined based on counting statistics and at higher temperatures are within the size of the symbols used.

The temperature dependence of  $F_P$  allows us to determine the barrier for the H<sub>2</sub>O recombination,  $\text{HO}_b + \text{HO}_t \rightarrow \text{H}_2\text{O}$ , by first-order kinetics (*SI Appendix, section S2*). With a prefactor of  $1 \times 10^{12} \text{ s}^{-1}$ , the optimum fit to the data (solid line, Fig. 2B) yields the recombination barrier,  $\Delta E_R = 0.355 \text{ eV}$ . The dotted lines show the lower- and upper-bound values of 0.345 and 0.365 eV, respectively. Further, assuming equilibrium at 140 K, the value of  $F_P(140\text{K}) = 0.05$  reveals that H<sub>2</sub>O is only 0.035 eV lower in free energy than the HO<sub>b</sub> + HO<sub>t</sub> pair.

In Fig. 3A, we further probe the H<sub>2</sub>O dissociation barrier,  $\text{H}_2\text{O} \rightarrow \text{HO}_b + \text{HO}_t$ , in experiments where different seeding gases (neat, He, H<sub>2</sub>) and nozzle temperatures are used to vary the H<sub>2</sub>O incident energy (see *Methods* and further details in *SI Appendix, section 3 and Fig. S5*). At low-incident energies ( $< 0.3 \text{ eV}$ ), the  $F_P$  is approximately constant ( $\sim 0.05$ ), increases above  $\sim 0.3 \text{ eV}$ , and reaches  $\sim 0.35$  at 1.3 eV. The observed break at  $\sim 0.3 \text{ eV}$  is a consequence of H<sub>2</sub>O dissociation barrier as discussed below. Similar  $F_P$  dependence is observed for the azimuth parallel to the Ti<sub>5c</sub> rows (*SI Appendix, Fig. S6*). This similarity along the two azimuths with very different corrugation of both physical and potential energy surfaces suggests that the energy dissipation may not be strongly dependent on the incident angle and exhibit so-called total energy scaling as seen previously on corrugated MgO(001) (32).

To understand the origin of the observed incident energy and temperature dependence of  $F_P$ , we perform ab initio molecular dynamics (AIMD) studies of H<sub>2</sub>O collisions with a stoichiometric TiO<sub>2</sub>(110). We note that extensive conformational sampling of molecularly bound H<sub>2</sub>O resulted in only a single stable binding configuration indicating that the P features are not the

result of an alternate metastable binding configuration. We start with a sample (100–200) of thermalized water configurations ( $T = 300 \text{ K}$ ) uniformly distributed at  $\sim 7 \text{ \AA}$  above the surface (*SI Appendix, Fig. S7*). We add translational kinetic energy,  $E_K = 0.1, 0.4, 0.7,$  and  $1.3 \text{ eV}$  with velocities in the direction concurrent with the experiment. We propagate each trajectory until the molecules collide with the surface and dissipate their excess energy; see *SI Appendix, section S5* for details. This procedure reproduces both the types and relative populations of species observed in the experiments, as reported in Fig. 3A (red squares) and *SI Appendix, Table S1*.

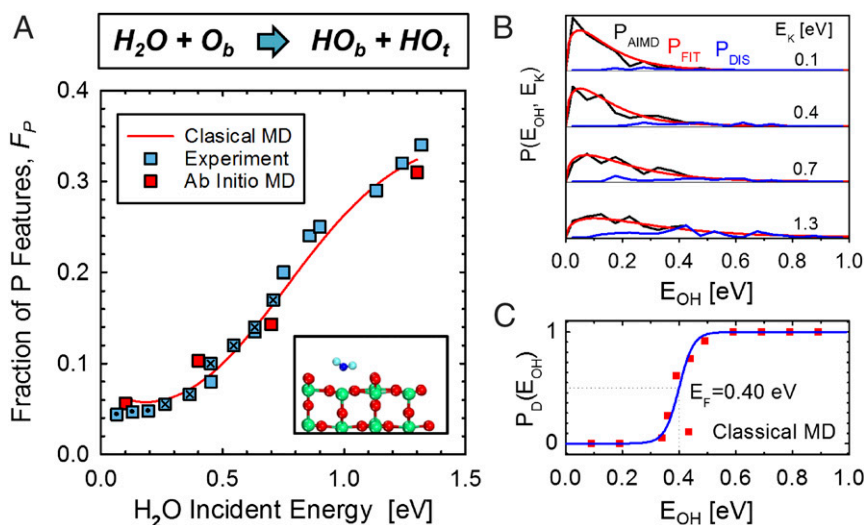
From the AIMD trajectories we observe the following: (i) At all incident energies, the fate of the species is determined by its collision point: those that land on the Ti<sub>5c</sub> sites dissociate and remain adsorbed, whereas the rest are scattered. In general, at low  $E_K$  there is a pronounced steering toward the Ti<sub>5c</sub> sites due to electrostatic attraction; hence, the overall number of retained species is highest at low energies. (ii) Irrespective of their fate after the collision, all of the molecules experience a reorientation as they approach the surface such that one OH bond is approximately perpendicular to the surface normal (Fig. 3A, *Inset*). This reorientation is the result of large, nonuniform electrostatic fields radiating out from the surface, coupling to the OH-bond dipole at distances  $\sim 6 \text{ \AA}$ , that orient the molecules before impact. We note that this reorientation breaks the twofold degeneracy associated with the water binding configuration, as discussed in more detail in *SI Appendix, section S5*. (iii) The torquing of the molecules induces coupling of the translational/rotational/vibrational modes and amplifies the kinetic energy transfer to the water OH-bond-stretching mode. The number of events leading to bond breaking steadily increases with  $E_K$ , and those that do break have on average a higher OH-bond kinetic energy,  $E_{OH}$ . Even for the lowest  $E_K = 0.1 \text{ eV}$ , there is a finite dissociation probability ( $\sim 5\%$ ) as a consequence of the steering of molecules into the bound state.

This long-range electrostatic coupling seen here for an oxide is distinct from metals and has consequences for how one can extract meaningful chemistry from molecular beam data. To generalize the interpretation of the  $E_K$ -dependent product cross-section, we note that this energy transfer effectively raises the temperature of the OH mode, and the resulting velocity distribution at the collision point can be represented by a Boltzmann distribution. AIMD trajectories resulting in OH bond breaking contribute to the long tail of  $P(E_{OH}, E_K)$  (Fig. 3B, blue line). To assess the probability,  $P_D(E_{OH})$ , of whether or not a molecularly adsorbed species with  $E_{OH}$  will break, we fit the reaction coordinate (Fig. 2B) to a classical electrostatic potential (see *SI Appendix, section S6* for details) and simulate an ensemble of events. The resulting  $P_D(E_{OH})$  (Fig. 3C) is well-described by a Fermi function, as a result of the existence of well-defined energy barrier. We can combine these factors into a probability describing the OH-bond-breaking process:

$$P_D(E_K) = \int P(E_{OH}, E_K) * P_D(E_{OH}) dE_{OH}. \quad [1]$$

This model accurately reproduces the results of the AIMD trajectories as shown in Fig. 3 and *SI Appendix, Fig. S17*. This approach allows us to decouple the vibrational excitation probability,  $P(E_{OH}, E_K)$ , from the bond-dissociation event probability,  $P_D(E_{OH})$ . Whereas  $P(E_{OH}, E_K)$  depends solely on electrostatics and is well-described by density-functional theory (DFT), the latter,  $P_D(E_{OH})$ , is associated with the potential energy landscape. Therefore, we fit the experimental data using Eq. 1 and obtain a barrier height of  $E_D = 0.36 \pm 0.01 \text{ eV}$  (Fig. 3A, red line), only  $\sim 0.04 \text{ eV}$  lower than that obtained from DFT. Inclusion of quantum and isotopic effects is further discussed in *SI Appendix,*





**Fig. 3.** (A) Water incident energy dependent fraction of P features,  $F_P$ , determined experimentally (blue squares) and theoretically (red squares, DFT AIMD trajectories). Seeding-gas-dependent data are coded by the square fill (neat  $\text{H}_2\text{O}$ , dot; He-seeded, cross; and  $\text{H}_2$ -seeded, empty). The red line is the fit to Eq. 1 yielding the dissociation barrier of  $0.36 \pm 0.01$  eV. (Inset) Schematic shows that the water molecule is oriented with an OH bond parallel to the surface. (B) The distribution of OH-bond kinetic energies  $P(E_{\text{OH}}, E_k)$  at the collision obtained from AIMD (black line) and a fit to a Boltzmann distribution (red). Distribution for a subset of trajectories resulting in OH bond breaking (blue). (C) The probability of OH-bond dissociation  $P_D(E_{\text{OH}})$  as a function of OH-bond kinetic energy as estimated from an ensemble of classical MD trajectories on the DFT potential energy surface.

section S7, providing a DFT estimate of a zero-point energy-corrected barrier  $E_D = 0.24$  eV, which is below that of experiment by 0.12 eV. The analysis presented in *SI Appendix, section S7* implies that the system is near the cross-over between the classical and quantum regimes at the experimental temperatures.

The complete energy diagram describing the equilibrium of water deprotonation on  $\text{TiO}_2(110)$  determined from our studies is shown in Fig. 4. These unique, site-specific experiments provide a direct measure of the acid/base properties of water on an oxide surface. A combination of different computational approaches unravels the mechanism and underlying physical principles leading to water dissociation. We reveal the surprising role of long-range electrostatics in the energy transfer and redistribution before collision with the surface. We further distill a simple and meaningful theoretical framework for the interpretation of the kinetic energy dependence of scattering cross-sections. This study provides a blueprint for extracting chemical insight from supersonic beam studies of oxide surfaces and provides energetic quantification of important surface reactions relevant to catalytic materials and environmental processes.

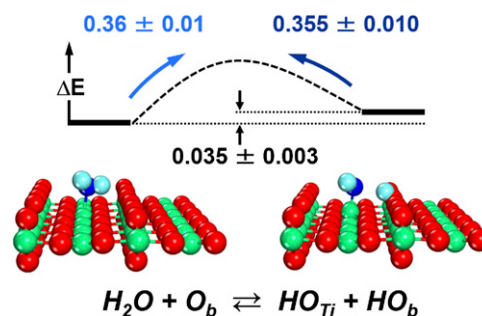
## Methods

**Experimental Setup.** Experiments were carried out in an Omicron low-temperature scanning tunneling microscope (LT-STM) system, which consists of a preparation chamber, microscopy chamber, and molecular beam chamber. The system is equipped with LT-STM/AFM, XPS, UV photoelectron spectroscopy, and low-energy electron diffraction. The  $\text{TiO}_2(110)$  sample (Princeton Scientific) was mounted on a standard Omicron Ta sample holder. The sample was cleaned by repeated cycles of  $\text{Ne}^+$  sputtering and annealing up to 900–950 K. Auger electron spectroscopy was used to determine the presence of impurities on the sample surface. The concentration of  $\text{V}_{\text{OS}}$  on the surface determined from the images was 15%. Electrochemically etched and UHV-annealed tungsten tips were used for imaging. In all STM experiments,  $\text{H}_2\text{O}$  coverage was kept very low ( $\sim 0.05$  ML) to keep the observed molecules isolated. All STM images were recorded in constant-current mode at a positive sample bias of 1.2–1.6 V and tunneling currents of 5–80 pA. Based on the information from the manufacturer, the reported  $\text{TiO}_2$  sample temperatures can be up to 3 K higher during counter heating ( $> 80$  K) than the real sample temperatures.

Deionized  $\text{H}_2\text{O}$  was purified by several freeze–pump–thaw cycles using liquid nitrogen and dosed on  $\text{TiO}_2(110)$  directly in the STM stage with the molecular beam incident at  $60^\circ$  with respect to the surface normal.

Experiments were carried out at different two azimuths pointing along and across the  $\text{Ti}_{5c}$  and  $\text{O}_b$  rows. Both quasi-effusive and supersonic molecular beams were used for water deposition. The neat  $\text{H}_2\text{O}$  beam was created by expanding 20 torr of water through a 50- $\mu\text{m}$  orifice. The supersonic beams were produced by passing 500 torr of either helium or hydrogen through a water bubbler at room temperature and expanding the resulting mixture through a 50- $\mu\text{m}$  orifice. A heatable nozzle assembly (300–900 K) was used to vary the energy of the incident  $\text{H}_2\text{O}$ . Further details are provided in *SI Appendix, section S3*.

**Computational Details.** All DFT calculations were performed using the CP2K package (33). The exchange-correlation energy was described by the generalized gradient approximation with the spin-polarized Perdew–Burke–Ernzerhof functional (34). The wave functions were expanded in optimized double- $\zeta$  Gaussian basis sets (35) and the plane waves were expanded with a cutoff energy of 400 Ry. Core electrons have been modeled by scalar relativistic norm-conserving pseudopotentials with 12, 4, and 1 valence electrons for Ti, O, and H, respectively (36). Brillouin zone integration is performed with a reciprocal space mesh consisting of only the  $\Gamma$ -points. The  $\text{TiO}_2(110)$ -p( $6 \times 2$ ) surface was used to model the  $\text{TiO}_2$  substrate, consisting of 6 O–Ti–O trilayers (18 atomic layers), and only the bottom Ti atomic layer was frozen while the remaining layers were allowed to relax. The slab was repeated periodically with a vacuum depth of  $\sim 20$  Å in the direction of the surface normal. The dependence of water dissociation equilibrium on the parameters and methods of the calculations has been discussed in detail elsewhere (13, 37).



**Fig. 4.** Values determined for the water deprotonation potential energy surface (in electron volts) on  $\text{TiO}_2(110)$ .

All AIMD simulations were performed by sampling initial conditions using a canonical [constant number, volume, temperature (NVT)] ensemble using Nosé–Hoover thermostats (38, 39) with a time step 0.5 fs. The collision simulations were performed within the microcanonical [constant number, volume, energy (NVE)] ensemble. The temperature of the TiO<sub>2</sub> substrate is set as 10 K and the initial translational kinetic energy of water,  $E_K$ , is chosen to be 0.1, 0.4, 0.7, and 1.3 eV, respectively, consistent with the range of beam energies in the experiments. To be comparable with the experiments, we started the simulation with a H<sub>2</sub>O molecule in the gas phase, initially located at  $\sim 7$  Å above the surface, with a velocity at 60° to the surface normal and perpendicular to the O<sub>b</sub> rows. We used 208 independent runs at  $E_K = -0.1$  eV due to the relatively low percentage of dissociated water seen in the experimental studies at low  $E_K$  values and 104 for  $E_K = 0.4, 0.7,$  and 1.3 eV; each run was propagated for a total time of 1–3 ps. A total of  $\sim 1$  ns of combined AIMD trajectories was performed and used for the data analysis presented in *SI Appendix, section S2*.

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